



# Photocatalytic water splitting over Pt-loaded TiO<sub>2</sub> (Pt/TiO<sub>2</sub>) catalysts prepared by the polygonal barrel-sputtering method

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## ABSTRACT

Pt-loaded TiO<sub>2</sub> (Pt/TiO<sub>2</sub>) catalysts for photocatalytic water splitting were prepared by the polygonal barrel-sputtering method. The results showed that the evolution rates of H<sub>2</sub> and O<sub>2</sub> gases for our Pt/TiO<sub>2</sub> catalysts were ca. 5–10 times of those prepared by a conventional photodeposition method. The molar ratios of H<sub>2</sub> versus O<sub>2</sub> were close to be 2, revealing that the stoichiometric water decomposition occurred. The characterizations of the prepared samples implied that the smaller Pt nanoparticles were highly dispersed on the TiO<sub>2</sub> particles by the polygonal barrel-sputtering method. The deposition of the smaller Pt particles increased the Pt surface areas, which can improve the water splitting activities. In addition, the particles deposited by the sputtering method were electron-deficient Pt metals that can enhance the water splitting reaction, whereas those by the photodeposition method were electron-rich Pt metals that reduce the effectiveness of the photocatalytic reaction. These different electron states would play an important role in the remarkable differences in the water splitting activities between the Pt/TiO<sub>2</sub> catalysts. Consequently, the polygonal barrel-sputtering method is useful for preparing photocatalysts with the high water splitting activities.

## 1. Introduction

Hydrogen (H<sub>2</sub>) is currently attracts considerable attentions as a renewable energy [1]. There are several methods for producing H<sub>2</sub> gas [2–15]. Among the available methods, photocatalytic water splitting is an environmentally friendly H<sub>2</sub> producing method, because of no emission of greenhouse gases such as carbon dioxide (CO<sub>2</sub>) [6–15].

In the photocatalytic water splitting, various kinds of semiconductor particles have been used [6–12]. When the semiconductors are irradiated, the holes and electrons are generated, leading to the following water splitting reactions [8]:



However, the water splitting over the semiconductors is difficult, since the holes and electrons generated recombine very easily [10].

On the other hand, nanoparticles such as metals and metal oxides can capture electrons generated from the semiconductors [12,13]. Therefore, numerous photocatalysts have been prepared by depositing nanoparticles on semiconductors in order to exclude the recombination of photoholes and electrons [12–24]. However, the water splitting performances of the current photocatalysts must be further improved to

achieve the highly efficient H<sub>2</sub> production.

It has been reported that the high dispersion of smaller nanoparticles on the semiconductors makes it possible to improve the water splitting performances of the catalysts [22,23]. Based on this finding, we focused our attention on our original particle surface modification process called as the “polygonal barrel-sputtering method” [25–39]. This method is based on a sputtering technique, in which nucleation and particle growth of the sputtered materials reportedly lead to the deposition of nanoparticles on supports [40]. For our method, the formation of the nanoparticles occurs on the particle surfaces used as the supports [39]. In addition, this original sputtering method contains no heating procedure, allowing the high dispersion of small nanoparticles of metals [32,34,35,38], metal alloys [31,33,36,37], or metal oxides [36]. In this study, Pt-loaded TiO<sub>2</sub> (Pt/TiO<sub>2</sub>) catalysts, which are well-known as a common photocatalyst for water splitting [14–19], were prepared by the polygonal-barrel sputtering method, and their photocatalytic water splitting performances were examined.

## 2. Experimental

### 2.1. Preparation of Pt/TiO<sub>2</sub> samples

The polygonal barrel-sputtering apparatus used for preparing Pt/

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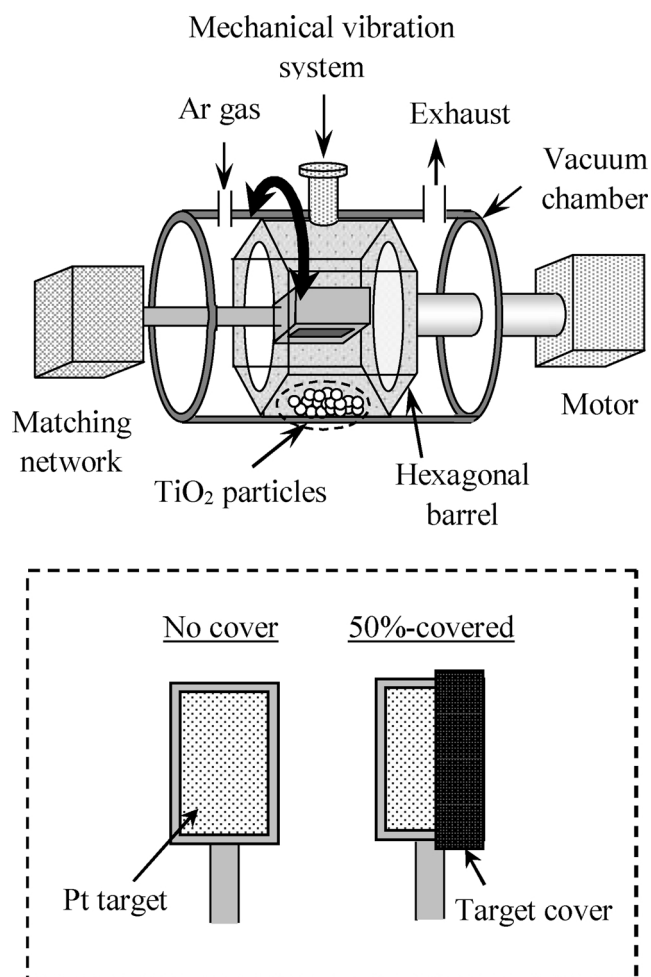


Fig. 1. Schematic diagram of the polygonal barrel-sputtering system.

TiO<sub>2</sub> samples is shown in Fig. 1 [25–39]. This apparatus is composed of a vacuum chamber including a hexagonal barrel, in which a powdered support is introduced. A Pt plate (99.99% purity, W 50 mm × 100 mm) was used as the target for sputtering. TiO<sub>2</sub> particles (mean primary particle size, 21 nm, P25; Nippon Aerosil) used as the support were heated at 180 °C to prevent aggregation of the particles by water during preparation in advance [31,35]. The Pt/TiO<sub>2</sub> sample was prepared as follows [25–39]. The pre-heated TiO<sub>2</sub> powder of 2 g was introduced into the hexagonal barrel and the barrel was placed in the vacuum chamber. The vacuum chamber was then carefully evacuated using a rotary pump and a diffusion pump. After the pressure was decreased to less than  $8 \times 10^{-4}$  Pa, Ar gas (purity: 99.9999%) was slowly introduced into the chamber. Subsequently, sputtering was performed at an Ar gas pressure of 0.8 Pa without heating. An AC power of 25 W was supplied by RF power generation (13.56 MHz) to highly disperse the small Pt nanoparticles on the surfaces of the TiO<sub>2</sub> particles. During the sputtering, to fragment the secondary particles of the TiO<sub>2</sub> into the primary particles, and then stir them, the hexagonal barrel was oscillated (intervals of 14 s and amplitude of  $\pm 75^\circ$ ) and vibrated mechanically. The amount of Pt deposited was controlled by changing the sputtering time between 5–15 min (see Table 1). When the samples with the Pt deposition amount of 0.05 and 0.10 wt.% were prepared, a half of the Pt target plate was covered with a stainless plate, as shown in Fig. 1. After sputtering, N<sub>2</sub> gas was gradually introduced to the atmospheric pressure, at which the prepared samples could be extracted. Hereafter, the samples prepared by the polygonal barrel-sputtering method are denoted as (BS)-Pt/TiO<sub>2</sub>.

Pt/TiO<sub>2</sub> samples used as references were prepared by a conventional

**Table 1**  
Sputtering conditions and Pt deposited amount of (BS)-Pt/TiO<sub>2</sub> samples.

Sputtering conditions <sup>a,b</sup>		Pt deposition amount <sup>c</sup> / wt. %
Covered area of Pt target	Sputtering time / min	
50%	5	0.05
50%	10	0.10
No cover	15	0.19

<sup>a</sup> The other sputtering conditions were Ar gas pressure of 0.8 Pa and AC power of 25 W.

<sup>b</sup> Sputtering was conducted without heating.

<sup>c</sup> The Pt deposited amounts were determined by ICP-AES.

photodeposition method using a closed gas-circulating system through the following procedure [16]. The TiO<sub>2</sub> of 1 g and H<sub>2</sub>PtCl<sub>6</sub>·6H<sub>2</sub>O (> 98.5% purity, Kanto Chemical) of a certain weight were uniformly mixed in a Pyrex glass-photoreactor including 1050 ml of aqueous methanol solution (10 vol.%) by stirring the solution at 500 rpm using a magnetic stirrer (MS 500D; Yamato). After the suspension was deaerated with Ar gas, the irradiation was conducted for 12 h using a 400 W-high-pressure Hg lamp (HL400B-8; Sen Tokushu Kogen) as the light source. During irradiation, the suspension was stirred at 500 rpm and the temperature was kept at 25 °C by circulating water through a cylindrical Pyrex jacket around the light source. The Pt/TiO<sub>2</sub> samples obtained by the above procedure are denoted as (PD)-Pt/TiO<sub>2</sub>.

## 2.2. Photocatalytic water splitting on Pt/TiO<sub>2</sub> [15–17]

Photocatalytic water splitting over the Pt/TiO<sub>2</sub> samples was performed in the same photoreactor as that used for the (PD)-Pt/TiO<sub>2</sub> preparation. The UV light irradiation was carried out while a Pt/TiO<sub>2</sub> sample of 0.3 g was suspended in 1050 ml of ion exchanged water (stirring velocity: 500 rpm). The reactant gas (1 ml) was periodically collected by a gas sampler, and it was analyzed by a gas chromatograph (GC2014; Shimadzu) equipped with a thermal conductivity detector (TCD) and a molecular sieve 5 A column (GL Science). In the gas chromatography, Ar gas used as the carrier gas was supplied at a rate of 50 sccm. The temperatures of the gas injector, column, and detector were set at 170, 150, and 170 °C, respectively.

## 2.3. Characterization of the prepared Pt/TiO<sub>2</sub> samples

The amounts of Pt deposited in the prepared samples were estimated by an inductively coupled plasma-atomic emission spectroscopy (ICP-AES: Optima 7300DV; PerkinElmer). The solutions for the ICP measurement were prepared by heating the Pt/TiO<sub>2</sub> sample (100 mg) in aqua regia (3 ml) for 1 h, followed by diluting the obtained solution to 100 ml with Millipore water. The microscopic properties of the Pt/TiO<sub>2</sub> samples were evaluated by transmission electron microscopy (TEM: JEM-2100; JEOL) at the accelerating voltage of 200 kV. The size distributions of the deposited Pt particles were obtained by directly measuring the size of more than 150 randomly chosen particles from the magnified TEM images. The prepared samples were also characterized by an X-ray diffraction analysis (XRD: Philips; PW1825/00) with Cu K $\alpha$  radiation operating at 40 kV and 30 mA. The optical properties of the Pt/TiO<sub>2</sub> samples were examined by an ultraviolet-visible (UV-VIS) spectrometer (V-560i RM; JASCO). The electron state of the deposited Pt was assessed by an X-ray photoelectron spectroscopy (XPS: ESCALAB 250X; Thermo Fisher Scientific) which was performed at 25 W using monochromatized Al K $\alpha$  excitation. The obtained spectra were referenced with respect to C1s core level of a bonding energy of 284.8 eV [41].

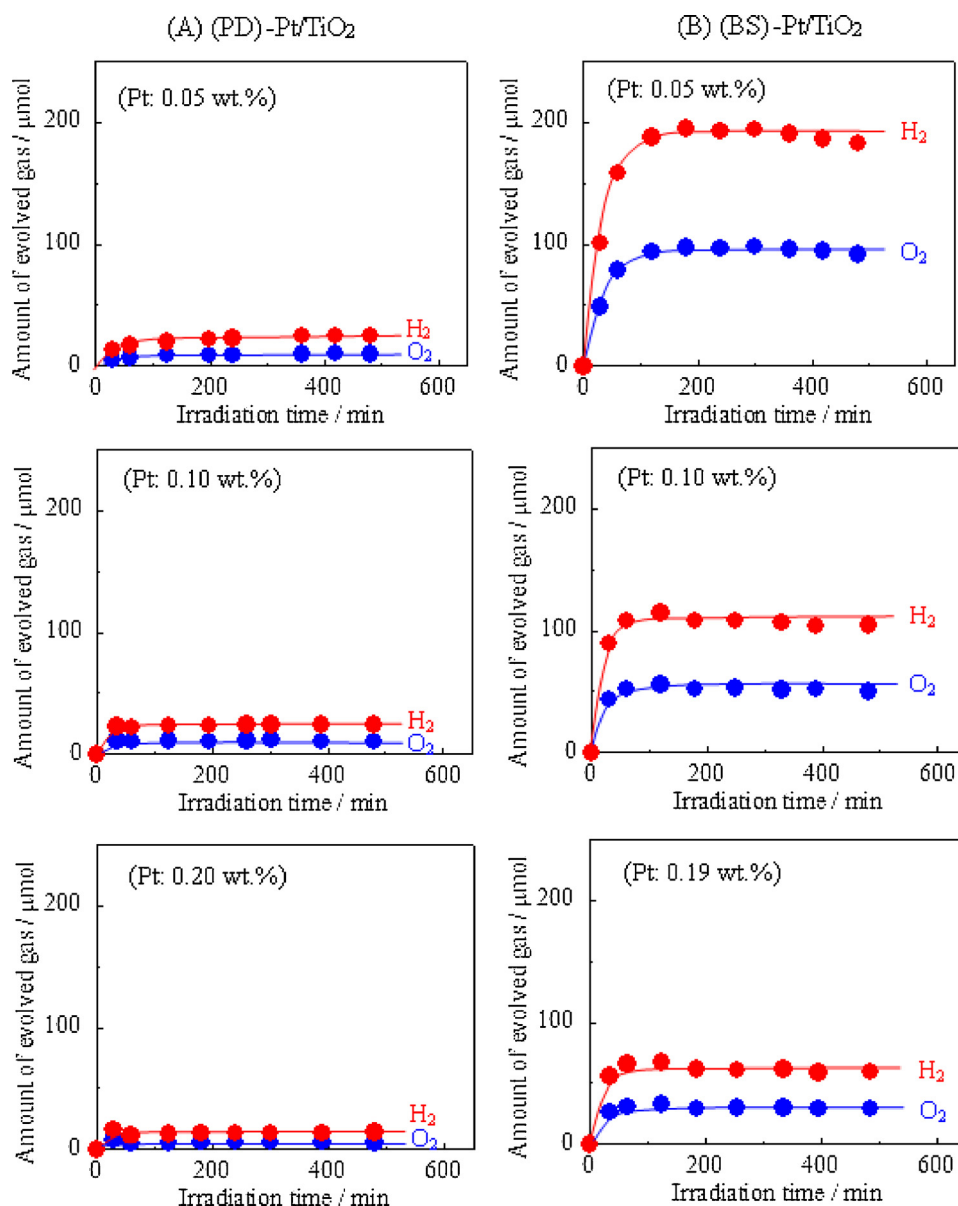


Fig. 2. Amounts of H<sub>2</sub> and O<sub>2</sub> gases generated by photocatalytic water splitting over (A) (PD)-Pt/TiO<sub>2</sub> and (B) (BS)-Pt/TiO<sub>2</sub> samples versus UV irradiation time.

Table 2

Evolution rates of H<sub>2</sub> and O<sub>2</sub> gases and Pt surface areas of Pt/TiO<sub>2</sub> samples used in photocatalytic water splitting.

Amount of deposited Pt / wt.%	Gas evolution rate / $\mu\text{mol h}^{-1}$			Pt surface area ( $\text{SA}_{\text{Pt}}$ ) <sup>a</sup> / $\text{cm}^2$	Turnover frequency / $\times 10^{-2} \text{ s}^{-1}$	
	H <sub>2</sub>	O <sub>2</sub>	H <sub>2</sub> /O <sub>2</sub> molar ratio		H <sub>2</sub>	O <sup>2</sup>
– (PD)-Pt/TiO <sub>2</sub> –						
0.05	17.1	7.3	2.4	144.7	4.80	2.05
0.10	22.0	10.5	2.1	190.7	4.70	2.24
0.20	11.2	4.6	2.4	349.7	1.30	0.53
– (BS)-Pt/TiO <sub>2</sub> –						
0.05	157.7	78.6	2.0	246.8	25.44	12.84
0.10	107.8	51.7	2.1	466.2	9.36	4.49
0.19	59.6	27.8	2.1	885.8	2.71	1.27

<sup>a</sup> The data were determined from Eq. (3).

### 3. Results and discussion

#### 3.1. Water splitting performances of the prepared Pt/TiO<sub>2</sub> catalysts

The photocatalytic water splitting over the Pt/TiO<sub>2</sub> samples was carried out in the ion exchanged water without any electrolytes. The H<sub>2</sub> and O<sub>2</sub> gases were evolved for all the samples, which might be attributed to the partial irradiation of the UV light from the top of the suspension [16,19]. Fig. 2 shows the amounts of the evolved H<sub>2</sub> and O<sub>2</sub> gases versus the irradiation time. Over (PD)-Pt/TiO<sub>2</sub>, the amounts of the H<sub>2</sub> and O<sub>2</sub> gases gradually increased until the irradiation time of 60 min (Fig. 2(A)). After 60 min, the gas evolution was suppressed. The factors inducing this trend are considered to be the increase in the pressure in the reactor due to the closed system [15] and the effect of the reverse reaction (H<sub>2</sub> + 1/2O<sub>2</sub> → H<sub>2</sub>O), as described later in this paper, although the details are uncertain. Similar trends of the gas evolutions were observed for the (BS)-Pt/TiO<sub>2</sub> samples, as shown in Fig. 2(B). However, the evolution amounts of H<sub>2</sub> and O<sub>2</sub> for (BS)-Pt/TiO<sub>2</sub> were significantly larger than those for (PD)-Pt/TiO<sub>2</sub>.

From the data at 60 min in Fig. 2, the evolution rates of H<sub>2</sub> and O<sub>2</sub>

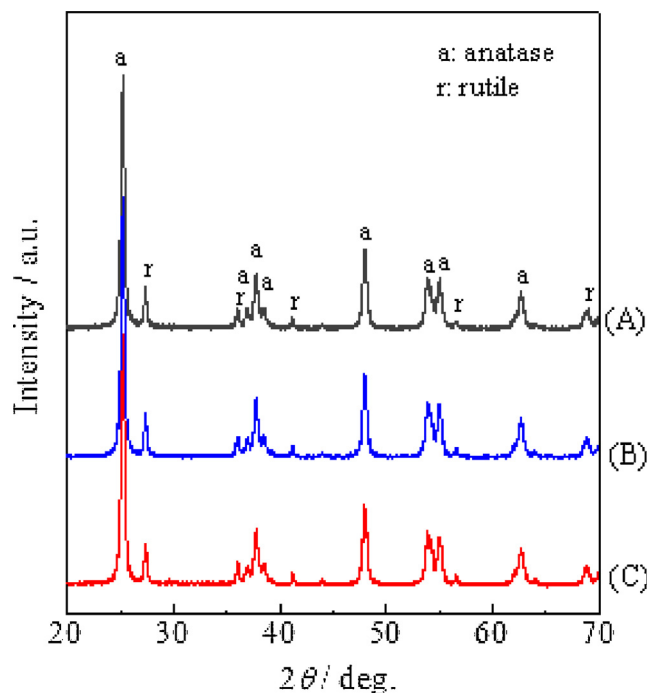


Fig. 3. XRD patterns of (A) as-received  $\text{TiO}_2$ , (B) (PD)-Pt/ $\text{TiO}_2$ , and (C) (BS)-Pt/ $\text{TiO}_2$ . The Pt deposition amounts for the Pt/ $\text{TiO}_2$  samples were 0.05 wt.%.

were determined. The obtained results are listed in Table 2. For (PD)-Pt/ $\text{TiO}_2$ , the evolution rates of  $\text{H}_2$  and  $\text{O}_2$  were 11.2–22.0 and 4.6–10.5  $\mu\text{mol h}^{-1}$ , respectively. In the case of (BS)-Pt/ $\text{TiO}_2$ , the gas evolution rates ( $\text{H}_2$ : 59.6–157.7  $\mu\text{mol h}^{-1}$ ,  $\text{O}_2$ : 27.8–78.6  $\mu\text{mol h}^{-1}$ ) were ca. 5–10 times greater than those for (PD)-Pt/ $\text{TiO}_2$  with the similar Pt deposition amounts. The water splitting activity of Pt/ $\text{TiO}_2$  was apparently improved by the polygonal barrel-sputtering method. It should be noted that the molar ratios of  $\text{H}_2$  versus  $\text{O}_2$  ( $\text{H}_2/\text{O}_2$  molar ratios) in Table 2 were close to 2 for all the samples, indicating that the gas evolutions occurred via the stoichiometric water decomposition of  $\text{H}_2\text{O} \rightarrow \text{H}_2 + 1/2\text{O}_2$ .

### 3.2. Factors inducing high water splitting activity of (BS)-Pt/ $\text{TiO}_2$

To investigate the factors of the high water splitting activities, the physical properties of the prepared samples were evaluated. Fig. 3 shows the XRD patterns of (A) as-received  $\text{TiO}_2$ , (B) (PD)-Pt/ $\text{TiO}_2$ , and (C) (BS)-Pt/ $\text{TiO}_2$ . Because of the small amount of the Pt deposited (= 0.05 wt.%), no peaks characteristic of Pt were observed for both Pt/ $\text{TiO}_2$  samples. All the diffraction peaks in the Pt/ $\text{TiO}_2$  samples agreed with the signals of the as-received  $\text{TiO}_2$  which were assigned to those of the anatase and rutile structures [42,43]. Thus, the crystal structure of  $\text{TiO}_2$  used as the supports did not change by the Pt deposition.

The UV-VIS spectra of the same samples are shown in Fig. 4. For (PD)-Pt/ $\text{TiO}_2$ , an absorption in the visible light region ( $> 400 \text{ nm}$ ) increased due to the fact that the incident photon frequency is resonant with the collective excitations of the conduction electrons of the deposited Pt [41]. Such absorption was not observed for (BS)-Pt/ $\text{TiO}_2$ , despite the identical Pt deposition amount to that for (PD)-Pt/ $\text{TiO}_2$ . The different absorption in the visible light region between (PD)-Pt/ $\text{TiO}_2$  and (BS)-Pt/ $\text{TiO}_2$  may concern with the sizes of the Pt particles in each sample, as described latter. It is noted that a strong absorption in the UV region ( $< 400 \text{ nm}$ ) observed for (BS)-Pt/ $\text{TiO}_2$  was similar to those for both (PD)-Pt/ $\text{TiO}_2$  and the  $\text{TiO}_2$  support, suggesting that the optical properties of the Pt/ $\text{TiO}_2$  samples did not affect the difference in their photocatalytic water splitting activities.

Fig. 5(A) and (B) show typical TEM images of (PD)-Pt/ $\text{TiO}_2$  and

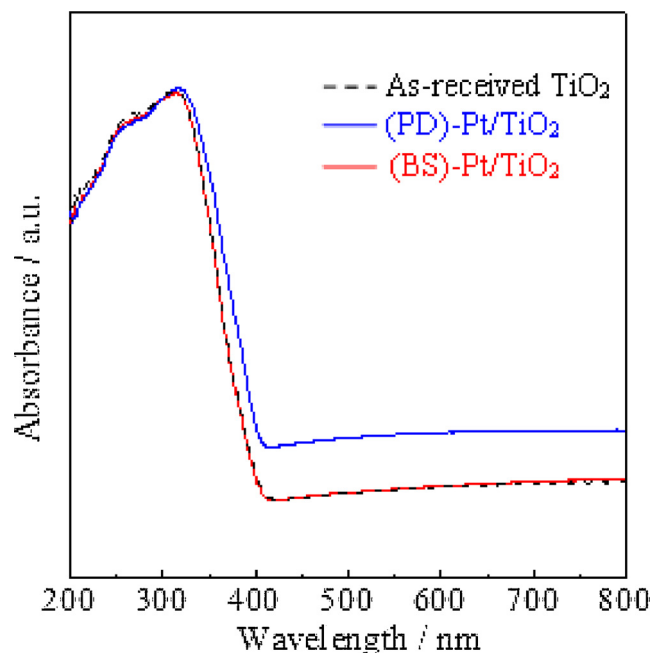


Fig. 4. UV-VIS spectra of as-received  $\text{TiO}_2$ , (PD)-Pt/ $\text{TiO}_2$ , and (BS)-Pt/ $\text{TiO}_2$ . The Pt deposition amounts for the Pt/ $\text{TiO}_2$  samples were 0.05 wt.%.

(BS)-Pt/ $\text{TiO}_2$ , respectively. For all the samples, Pt nanoparticles shown as black dots were observed on the gray  $\text{TiO}_2$  support background. As compared to (PD)-Pt/ $\text{TiO}_2$ , the Pt particles were highly dispersed in (BS)-Pt/ $\text{TiO}_2$ . The size distributions of Pt particles in each sample are also represented in Fig. 5. For (PD)-Pt/ $\text{TiO}_2$ , the Pt particles ranged from 1.0–4.6 nm at the Pt deposition amount of 0.05 wt.%. The particle size distribution for (PD)-Pt/ $\text{TiO}_2$  became wider by increasing the Pt deposition amounts to 0.10 and 0.20 wt. %. The center of the particle sizes also shifted to the larger size with the increase in the Pt deposition amount. From these size distributions, the mean particle sizes were estimated to be 2.9 nm (Pt: 0.05 wt.%), 4.4 nm (Pt: 0.10 wt.%), and 4.8 nm (Pt: 0.20 wt.%). As for (BS)-Pt/ $\text{TiO}_2$ , the Pt particles had similar sizes of 0.8–3.0 nm for all the samples. The mean particle sizes were ca. 1.8 nm, which was smaller than those of (PD)-Pt/ $\text{TiO}_2$ .

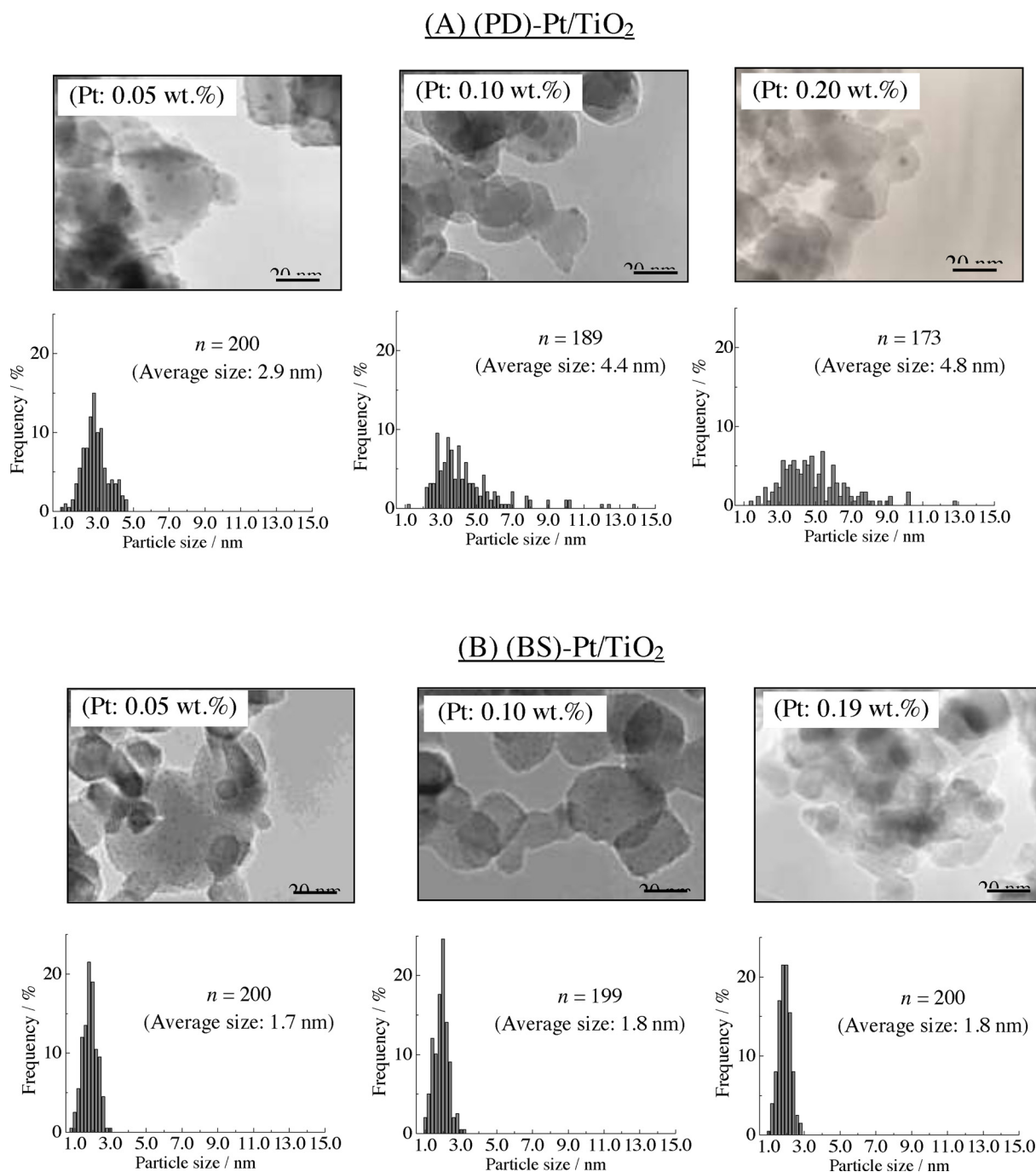
When the shapes of the deposited Pt particles are supposed to be sphere, the Pt surface areas of the samples used in the photocatalytic reactions ( $\text{SA}_{\text{Pt}}$ , in  $\text{cm}^2$ ) can be calculated using the following equation [44,45]:

$$\text{SA}_{\text{Pt}} = (6000/\rho d \times 10^4) \times w/100 \times 0.3 \quad (3)$$

where  $\rho$  is the Pt density ( $21.45 \text{ g cm}^{-3}$ ),  $d$  and  $w$  are mean particle size of the deposited Pt obtained by TEM (in nm) and its deposition amount (in wt.%), respectively, and 0.3 is a weight of the Pt/ $\text{TiO}_2$  sample used in the reaction. The  $\text{SA}_{\text{Pt}}$  values calculated from Eq. (3) are summarized in Table 2. The (BS)-Pt/ $\text{TiO}_2$  samples with the Pt deposition amounts of 0.05, 0.10, and 0.19 wt.% gave the  $\text{SA}_{\text{Pt}}$  values of 246.8, 466.2, and 885.8  $\text{cm}^2$ , respectively. These values were 1.7–2.5 times of those of the (PD)-Pt/ $\text{TiO}_2$  samples with the similar Pt deposition amounts, which can contribute to the high water splitting activities of (BS)-Pt/ $\text{TiO}_2$ . However, the turnover frequencies of the  $\text{H}_2$  and  $\text{O}_2$  evolution for (BS)-Pt/ $\text{TiO}_2$ , which were semi-quantitatively estimated using the  $\text{SA}_{\text{Pt}}$  values and the atomic radius of Pt (1.39 Å), were 2.0–5.3 times larger than those for (PD)-Pt/ $\text{TiO}_2$  (see Table 2). In addition, the turnover frequencies obtained for (BS)-Pt/ $\text{TiO}_2$  decreased with the increasing Pt deposited amount. These results suggest that there were other factors affecting the water splitting activity of Pt/ $\text{TiO}_2$ .

The distances between the Pt particles were also estimated from the TEM images. As a result, the mean distances between the Pt particles deposited in (BS)-Pt/ $\text{TiO}_2$  decreased from 8.4–6.1 and 5.3 nm as the Pt





**Fig. 5.** Typical TEM images (accelerating voltage, 200 kV) and particle size distributions ( $n$ : number of particle counts) of (A) (PD)-Pt/TiO<sub>2</sub> and (B) (BS)-Pt/TiO<sub>2</sub> samples.

deposition amount was increased from 0.05 to 0.10 and 0.19 wt.%. For the water splitting reaction, the H<sub>2</sub> and O<sub>2</sub> generated tend to convert again into water on the Pt particles via a reaction of  $\text{H}_2 + 1/2\text{O}_2 \rightarrow \text{H}_2\text{O}$  [17,18]. Indeed, even over (BS)-Pt/TiO<sub>2</sub> with the Pt deposition amount of 0.05 wt.%, the reverse reaction of  $\text{H}_2 + 1/2\text{O}_2 \rightarrow \text{H}_2\text{O}$  occurred at the rates of  $\text{H}_2 = 14.1 \mu\text{mol/h}$  and  $\text{O}_2 = 7.8 \mu\text{mol/h}$  when the UV lamp was turned off after the H<sub>2</sub> and O<sub>2</sub> evolution became steady-state. The frequency of this reverse reaction should increase by the approach of the Pt particles. Moreover, the reverse reaction might contribute to the suppression of the H<sub>2</sub> and O<sub>2</sub> evolution after 60 min. On the other hand, the decrease in the distances between the Pt particles also means the increase in the Pt coverage on the TiO<sub>2</sub> surfaces. This factor induces the inner filter effect, which suppresses the water

splitting reaction [23]. Based on these factors, the difference in the turnover frequencies for (BS)-Pt/TiO<sub>2</sub> in Table 2 can be well explained by the Pt particle distances. However, for (PD)-Pt/TiO<sub>2</sub>, the mean distance between the Pt particles was estimated to be 10.1 nm at the Pt deposition amount of 0.05 wt.%, which was longer than that of the (BS)-Pt/TiO<sub>2</sub> with the same Pt amount, despite the low water splitting activity. It is thereby considered that the distances between the Pt particles did not originally participate in the different water splitting activities between (BS)-Pt/TiO<sub>2</sub> and (PD)-Pt/TiO<sub>2</sub>.

The electron states of the Pt deposited were examined by XPS. Fig. 6 shows the Pt4f spectra of the (PD)-Pt/TiO<sub>2</sub> and (BS)-Pt/TiO<sub>2</sub> samples. It is seen in the spectra of the (PD)-Pt/TiO<sub>2</sub> samples that two deconvoluted peaks corresponding to the Pt4f<sub>7/2</sub> and Pt4f<sub>5/2</sub> peaks were

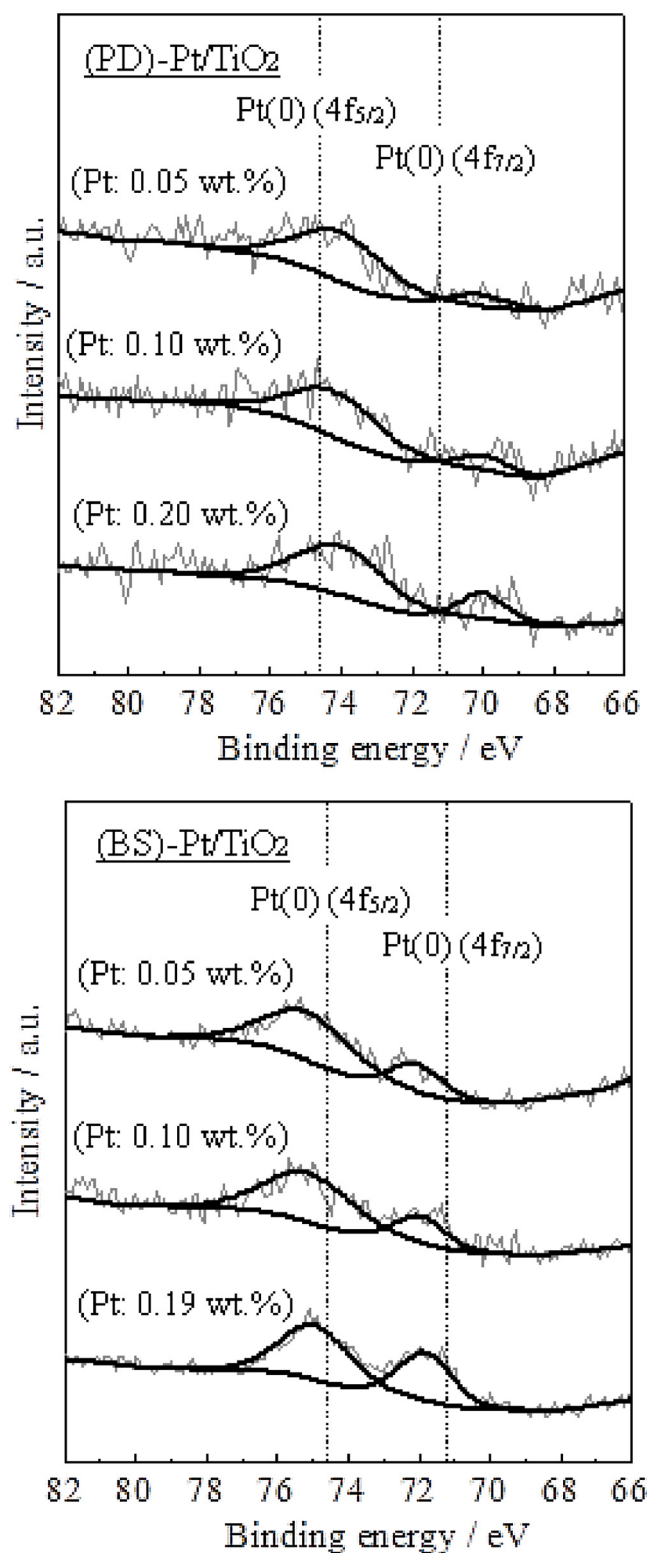


Fig. 6. Pt4f spectra of (PD)-Pt/TiO<sub>2</sub> and (BS)-Pt/TiO<sub>2</sub> obtained by XPS. The dashed lines show typical binding energies of Pt4f<sub>5/2</sub> and Pt4f<sub>7/2</sub> peaks for metallic Pt [43].

observed at 70.1 and 74.0 eV (Pt: 0.05 wt.%), 70.0 and 74.1 eV (Pt: 0.10 wt.%), and 70.0 and 74.0 eV (Pt: 0.20 wt.%). As compared to the reported peak positions of a metallic Pt expressed by dashed lines in this figure (71.2 and 74.6 eV [46]), the Pt4f peaks for (PD)-Pt/TiO<sub>2</sub> apparently shifted to the lower binding energies. The peak shifts to the lower binding energy indicate the deposition of the Pt metal particles with an

electron-rich state [47,48]. For the photodeposition, defect sites (TiO<sub>2</sub>-δ) included in TiO<sub>2</sub> plays an important role as the active sites for the formation of the metal particles [48]: Ti<sup>3+</sup> in TiO<sub>2</sub>-δ, which acts as the electron trap sites, leads to the electron transfer from TiO<sub>2</sub> to Pt, resulting in the deposition of the electron-rich Pt metal particles [47,48]. It has also been reported that the electron-rich Pt metals may repel electrons, which reduces the effectiveness of photocatalytic reaction [49]. This negative effect most likely contributed to the low water splitting activities of the (PD)-Pt/TiO<sub>2</sub> samples.

As for the (BS)-Pt/TiO<sub>2</sub> samples, the deconvoluted Pt4f<sub>7/2</sub> and Pt4f<sub>5/2</sub> peaks were observed at 72.1 and 75.1 eV (Pt: 0.05 wt.%), 72.0 and 75.1 eV (Pt: 0.10 wt.%), and 71.8 and 74.9 eV (Pt: 0.19 wt.%). The Pt4f peak positions for (BS)-Pt/TiO<sub>2</sub>, as mentioned above, were higher than those of the metallic Pt, while those were lower than those reported for Pt(II) (72.8–73.1 and 76.3–76.4 eV [50]) and Pt(IV) (74.6–74.9 and 78.1–78.2 eV [50]). These results are completely different from those for (PD)-Pt/TiO<sub>2</sub> and show that the Pt particles in (BS)-Pt/TiO<sub>2</sub> were composed of the Pt metals with an electron-deficient state that can enhance the electron collection [23,51,52]: this factor positively affects the photocatalytic reaction [23,51,52], which is probably induced by the initial state effects, e.g. charge transfer from metals to supports, lattice strain due to the small particle sizes, existence of many under-coordinated atoms, and so on [53,54]. In addition, similar peak shift was observed for (BS)-Pt/TiO<sub>2</sub> with Pt = 0.05 wt.% after the photoreaction, suggesting that the electron-deficient state of Pt was remained during the reaction. Consequently, the deposition of the electron-deficient Pt particles is most likely to originally contribute to the high water splitting activities of (BS)-Pt/TiO<sub>2</sub>. It is emphasized that the XPS peaks of metals such as Au, Ag, Ni, Cu, and Pt have been reported to shift to the higher binding energies with the decrease in the particle sizes [53–58]. Especially, the study reported by Garbarino et al. has shown that the significantly positive shifts of the Pt4f peaks were observed at the particle sizes of less than 3 nm [54]. Based on these reports, the deposition of the electron-deficient Pt particles for (BS)-Pt/TiO<sub>2</sub> may be attributed to the Pt particle sizes of less than 3 nm, as described in Fig. 5.

#### 4. Conclusions

In this study, we prepared the Pt/TiO<sub>2</sub> catalysts by the polygonal barrel-sputtering method and their photocatalytic water splitting performances were compared to those prepared by the conventional photodeposition method. The obtained results are summarized as follows.

- (1) For the photocatalytic water splitting over the (BS)-Pt/TiO<sub>2</sub> catalysts, H<sub>2</sub> and O<sub>2</sub> were evolved via the reaction of H<sub>2</sub>O → H<sub>2</sub> + 1/2O<sub>2</sub>. The gas evolution rates were ca. 5–10 times larger than those over the (PD)-Pt/TiO<sub>2</sub> catalysts.
- (2) The TEM measurements showed that the deposited Pt particles were highly dispersed on the TiO<sub>2</sub> powder support by the polygonal barrel-sputtering method. The sizes of the deposited particles were smaller than those in the sample prepared by the photodeposition method. The deposition of the small Pt particles increased the Pt surface area, which contributed to the high water splitting activities of (BS)-Pt/TiO<sub>2</sub>.
- (3) The Pt particles deposited in (PD)-Pt/TiO<sub>2</sub> were the electron-rich Pt metals, while those in (BS)-Pt/TiO<sub>2</sub> were the electron-deficient Pt metals. These different electron states of the deposited Pt would mainly affect the water splitting activities of the Pt/TiO<sub>2</sub> catalysts.

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